

CMo Structure: AB_hP12_194_af_bf-001

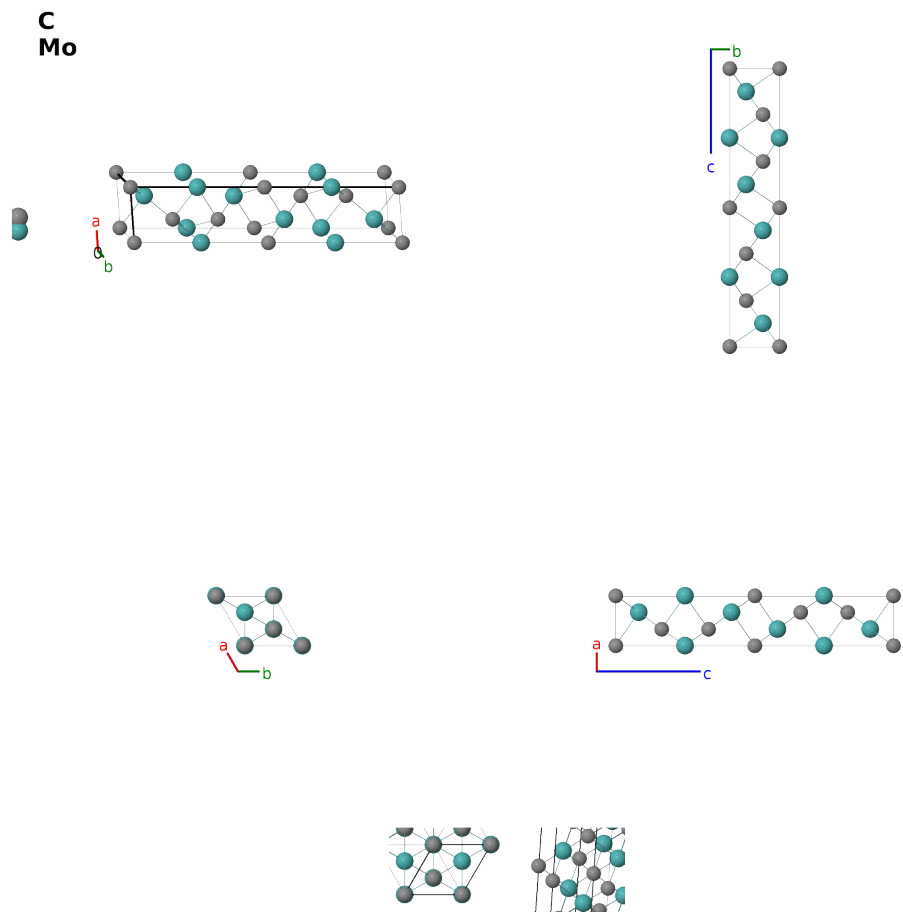
This structure previously used the label(s) **AB_hP12_194_af_bf**. Calls to these addresses will be redirected here.

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<https://aflow.org/prototype-encyclopedia/Q5G8>

https://aflow.org/prototype-encyclopedia/AB_hP12_194_af_bf-001



Prototype	CMo
AFLOW prototype label	AB_hP12_194_af_bf-001
ICSD	44987
CCDC	1614303
Pearson symbol	hP12
Space group number	194
Space group symbol	$P6_3/mmc$

AFLOW prototype command `aflow --proto=AB_hP12_194_af_bf-001`
 `--params=a, c/a, z3, z4`

Other compounds with this structure

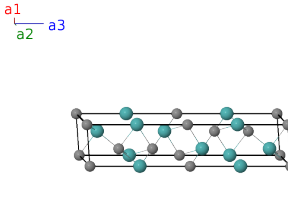
CRe, BaGaGe, CaGaGe, CaGaSn, CeNiP, DyNiP, GaGeSr, GdNiP, HoNiP, LiNiP, LuNiP, LuNiP, NdNiP, NiPPr, NiPTb, NiPTm, NiPY, C₂GeTi₃, C₂SiTi₃, C₂AlTi₃

- Note that all of the atoms sit on close packed <0001> planes. The stacking sequence may be written:

Atom	Mo-II	C-II	C-I	C-II	Mo-II	Mo-I	Mo-II	C-II	C-I	C-II	Mo-II	Mo-I
Position	B	C	A	B	C	A	C	B	A	C	B	A

- Thus the Mo-II atoms and all of the C atoms are always in an fcc-like local environment, while the Mo-I atoms are in an hcp-like local environment. Like AlN₃Ti₄, this is a MAX phase. For more information, see (Radovic, 2013).

Hexagonal primitive vectors

$$\begin{aligned}
 \mathbf{a}_1 &= \frac{1}{2}a \hat{\mathbf{x}} - \frac{\sqrt{3}}{2}a \hat{\mathbf{y}} \\
 \mathbf{a}_2 &= \frac{1}{2}a \hat{\mathbf{x}} + \frac{\sqrt{3}}{2}a \hat{\mathbf{y}} \\
 \mathbf{a}_3 &= c \hat{\mathbf{z}}
 \end{aligned}$$


Basis vectors

	Lattice coordinates		Cartesian coordinates	Wyckoff position	Atom type
$\mathbf{B}_1$	= 0	=	0	(2a)	C I
$\mathbf{B}_2$	= $\frac{1}{2} \mathbf{a}_3$	=	$\frac{1}{2}c \hat{\mathbf{z}}$	(2a)	C I
$\mathbf{B}_3$	= $\frac{1}{4} \mathbf{a}_3$	=	$\frac{1}{4}c \hat{\mathbf{z}}$	(2b)	Mo I
$\mathbf{B}_4$	= $\frac{3}{4} \mathbf{a}_3$	=	$\frac{3}{4}c \hat{\mathbf{z}}$	(2b)	Mo I
$\mathbf{B}_5$	= $\frac{1}{3} \mathbf{a}_1 + \frac{2}{3} \mathbf{a}_2 + z_3 \mathbf{a}_3$	=	$\frac{1}{2}a \hat{\mathbf{x}} + \frac{\sqrt{3}}{6}a \hat{\mathbf{y}} + cz_3 \hat{\mathbf{z}}$	(4f)	C II
$\mathbf{B}_6$	= $\frac{2}{3} \mathbf{a}_1 + \frac{1}{3} \mathbf{a}_2 + (z_3 + \frac{1}{2}) \mathbf{a}_3$	=	$\frac{1}{2}a \hat{\mathbf{x}} - \frac{\sqrt{3}}{6}a \hat{\mathbf{y}} + c(z_3 + \frac{1}{2}) \hat{\mathbf{z}}$	(4f)	C II
$\mathbf{B}_7$	= $\frac{2}{3} \mathbf{a}_1 + \frac{1}{3} \mathbf{a}_2 - z_3 \mathbf{a}_3$	=	$\frac{1}{2}a \hat{\mathbf{x}} - \frac{\sqrt{3}}{6}a \hat{\mathbf{y}} - cz_3 \hat{\mathbf{z}}$	(4f)	C II
$\mathbf{B}_8$	= $\frac{1}{3} \mathbf{a}_1 + \frac{2}{3} \mathbf{a}_2 - (z_3 - \frac{1}{2}) \mathbf{a}_3$	=	$\frac{1}{2}a \hat{\mathbf{x}} + \frac{\sqrt{3}}{6}a \hat{\mathbf{y}} - c(z_3 - \frac{1}{2}) \hat{\mathbf{z}}$	(4f)	C II
$\mathbf{B}_9$	= $\frac{1}{3} \mathbf{a}_1 + \frac{2}{3} \mathbf{a}_2 + z_4 \mathbf{a}_3$	=	$\frac{1}{2}a \hat{\mathbf{x}} + \frac{\sqrt{3}}{6}a \hat{\mathbf{y}} + cz_4 \hat{\mathbf{z}}$	(4f)	Mo II
$\mathbf{B}_{10}$	= $\frac{2}{3} \mathbf{a}_1 + \frac{1}{3} \mathbf{a}_2 + (z_4 + \frac{1}{2}) \mathbf{a}_3$	=	$\frac{1}{2}a \hat{\mathbf{x}} - \frac{\sqrt{3}}{6}a \hat{\mathbf{y}} + c(z_4 + \frac{1}{2}) \hat{\mathbf{z}}$	(4f)	Mo II
$\mathbf{B}_{11}$	= $\frac{2}{3} \mathbf{a}_1 + \frac{1}{3} \mathbf{a}_2 - z_4 \mathbf{a}_3$	=	$\frac{1}{2}a \hat{\mathbf{x}} - \frac{\sqrt{3}}{6}a \hat{\mathbf{y}} - cz_4 \hat{\mathbf{z}}$	(4f)	Mo II
$\mathbf{B}_{12}$	= $\frac{1}{3} \mathbf{a}_1 + \frac{2}{3} \mathbf{a}_2 - (z_4 - \frac{1}{2}) \mathbf{a}_3$	=	$\frac{1}{2}a \hat{\mathbf{x}} + \frac{\sqrt{3}}{6}a \hat{\mathbf{y}} - c(z_4 - \frac{1}{2}) \hat{\mathbf{z}}$	(4f)	Mo II

References

- [1] H. Nowotny, R. Parthé, and F. Benesovsky, *Das Dreistoffsystm: Molybdän-Silizium-Kohlenstoff*, Monatsh. Chem. Verw. Tl. **85**, 255–272 (1954).
- [2] M. Radovic and M. W. Barsoum, *MAX phases: Bridging the gap between metals and ceramics*, American Ceramic Society Bulletin **92**, 20–27 (2013).

First cited in

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